

Materials Science inc. Nanomaterials & Polymers

Synthesis and Properties of Au Hydride

Devika Sil⁺,^[a] Christopher Lane⁺,^[b] Ethan Glor⁺,^[c] Kyle D. Gilroy,^[d, e] Safiya Sylla,^[a] Bernardo Barbiellini,^[b, f] Robert Markiewicz,^[b] Maryam Hajfathalian,^[e] Svetlana Neretina,^[h] Arun Bansil,^[b] Zahra Fakhraai,^[c] and Eric Borguet^{*[a]}

The generation of chemical species from gases, noble metals and light interacting with localized surface plasmons represents a new paradigm for achieving low energy sustainable reaction pathways. Here, we demonstrate that the dissociation reaction of H₂ mediated by the decay of localized surface plasmons of gold nanoparticles leads to the generation of a new material as detected by a change in the optical properties of the gold nanostructures. The effective permittivity measured by *in situ* spectroscopic ellipsometry shows a blue-shift of 0.02 eV in the surface plasmon resonance, demonstrating the plausible formation of a metastable gold hydride layer on the surface of nanoparticles following the dissociation of H₂. The formation of this gold hydride through the interaction of gold with atomic H is supported by first-principles simulations. These calculations do not indicate a significant charge transfer upon hydrogenation of the (111) surface but rather large Friedel charge oscillations within the gold layer. Moreover, our blue-shift is produced by the formation of a hydride leading to changes in critical band gaps in the electronic structure. For a coverage of 11%, the calculated peak of the imaginary part of the ZZ-component of the dielectric tensor undergoes a blue shift of 28 nm from a hydrogen free peak at 574 nm.

Introduction

The interaction of gold with hydrogen, especially the formation of Au–H bonds, remains poorly understood due to the inert nature of bulk gold. According to Mukherjee *et al.*^[1] thermodynamics alone cannot drive the dissociation of H₂ given the large activation energy of 4.51 eV. Therefore pathways to gold-hydride formation remain elusive, except possibly in some extreme conditions.^[2,3] The interaction of noble metals with photons whose energy is in resonance with the surface plasmons leads to the generation of hot electrons that can drive chemical reactions with a relatively low energy cost. In fact, unlike experimental methods of high-pressure and high-temperature synthesis of metal hydrides,^[3] plasmon driven syntheses operate under gentle ambient conditions. Clearly, noble metal nanoparticles can trigger reactions in which they provide both hot electrons via the localized surface plasmon resonance and the active site for photocatalysis.^[4] These systems are being deployed in an increasing number of research fields including water splitting,^[5,6] solar energy harvesting,^[7] sensing,^[8,9] catalysis,^[1,10,11] generation of H₂ from alcohol,^[12] hydrocarbon conversion,^[13] fabrication of novel molecular electronic devices,^[14] plasmonic switches,^[15] quantum-dot plasmonic hybrid systems^[16–18] and nanoscale electronics.^[19]

Mukherjee *et al.*^[1] provided the first experimental evidence for the room-temperature photocatalytic dissociation of H₂ on gold nanoparticles using visible light by observing the production of HD from a mixture of H₂ and D₂. They inferred that a fraction of the hot electrons, generated by the resonantly excited gold nanoparticles, transfer into the antibonding orbitals of a physisorbed H₂ molecule, which then leads to the dissociation of the H₂ molecule. Mukherjee *et al.*^[1] thus showed that the localized surface plasmon resonance in gold can dissociate H₂ without the need for an active material such as Pd. Sil *et al.*^[20] leveraged this dissociation process to develop an all optical H₂ sensing scheme by exploiting shifts in the dielectric function of gold nanoparticle thin films upon adsorption of hydrogen.^[21] Related studies have observed similar shifts, however there is no consensus on the direction of the shift, blue or red. In this study, we show how the formation of a metastable gold hydride layer hypothesized by Sil *et al.*,^[20] Silverwood *et al.*,^[22] Whittaker *et al.*^[23] and Ishida *et al.*,^[24,25] following H₂ dissociation, explains the blue shift in the dielectric function of the gold-nanoparticle surface. *In situ* spectroscopic ellipsometry combined with state-of-the-art density

[a] Dr. D. Sil,⁺ S. Sylla, Prof. E. Borguet
Department of Chemistry, Temple University, Philadelphia, Pennsylvania 19122, USA.

E-mail: eborguet@temple.edu

[b] C. Lane,⁺ Prof. B. Barbiellini, Prof. R. Markiewicz, Prof. A. Bansil
Northeastern University Physics Department, 360 Huntington Ave. 111 Dana Research Center, Boston, MA 02115, USA.

[c] Dr. E. Glor,⁺ Prof. Z. Fakhraai
Department of Chemistry, University of Pennsylvania, 231 S. 34th Street, Philadelphia, PA 19104, USA.

[d] Dr. K. D. Gilroy
The Wallace H. Coulter Department of Biomedical Engineering, Georgia Institute of Technology and Emory University, Atlanta, GA 30332, USA.

[e] Dr. K. D. Gilroy, Dr. M. Hajfathalian
College of Engineering, Temple University, Philadelphia, PA 19122, USA

[f] Prof. B. Barbiellini
Department of Physics, School of Engineering Science, LUT University, FI-53850 Lappeenranta, Finland

[h] Prof. S. Neretina
College of Engineering, University of Notre Dame, Notre Dame, IN 46556, USA

[⁺] These authors contributed equally to this work

Supporting information for this article is available on the WWW under <https://doi.org/10.1002/slct.201900925>

functional theory (DFT) based first-principles computations justify our proposed mechanism based on gold hydride formation and elicit unique insights into the interaction of gold nanoparticles with atomic hydrogen.

Results and discussion

Measurements were performed using spectroscopic ellipsometry to investigate the effect of hydrogen on the dielectric function of dewetted gold nanoparticle thin film samples on microscope glass slides under broadband illumination. Sample preparation details are given in the Methods section.

Figure 1a illustrates raw ellipsometry data including the amplitude component (Ψ) and the phase difference (Δ). The

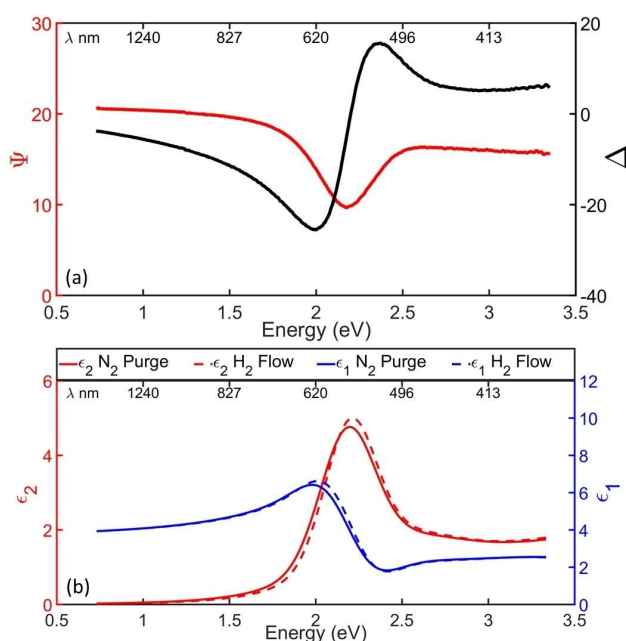


Figure 1. Raw ellipsometry data and fit used to obtain the dielectric function. (a) Raw ellipsometry data Ψ (red) and Δ (black) as a function of frequency and wavelength for a film held under N_2 conditions. The fit cannot be distinguished from the data. (b) Real (ϵ_1 , blue curves) and imaginary (ϵ_2 , red curves) parts of the dielectric function calculated based on the ellipsometry fits for samples held under N_2 (solid) and H_2 (dashed). All experiments carried out under broadband illumination.

real (ϵ_1) and imaginary (ϵ_2) parts of the measured dielectric function are shown in Figure 1b before and after exposure to H_2 . The peak location in frequency of ϵ_2 was used to monitor changes in time of the film's surface. The time sequence of exposures to N_2 or H_2 is illustrated in Figure 2a. The process can be divided into several steps. Initially, a dewetted gold nanoparticle sample, characterized by AFM^[26] and shown in Figure 2c, was exposed to nitrogen at 100 sccm under broadband illumination for 34 min. Then, the sample was exposed to a mixture of 95% nitrogen and 5% hydrogen (100 sccm) for 27 min, also under broadband illumination. During the H_2 flow, the ϵ_2 peak blue shifted from a frequency of 2.202 eV (563 nm)

to 2.217 eV (559 nm) as shown in Figure 1b. After hydrogen flow, nitrogen was turned on again for 30 min but recovery to the original state was not observed. Therefore, the top cover of the cell was taken off to further help purge the cell and nitrogen was kept on for an additional 35 min. This final step provided the recovery to the initial signal level.

Clearly, an overall 0.02 eV blue shift in ϵ_2 was detected during H_2 exposure. Incidentally, a similar blue shift upon hydrogenation of gold nanoparticles in a H_2 electrical discharge was reported by Giangregorio *et al.*,^[2] and a blue shift in the LSPR of Au nanorods was observed in the presence of H_2 and Pt nanoparticles, presumably the source of atomic H by spillover.^[27] Therefore, the presence of hydrogen significantly modifies the dielectric function, but nitrogen alone does not induce significant modifications. In fact, changes produced by nitrogen were smaller than 1% whereas the corresponding changes for hydrogen were as large as 4%. Moreover, switching the atmosphere from nitrogen to argon produced the same effect.^[26]

In order to interpret experimental results, guide analysis and serve as a baseline, first-principles, DFT-based calculations were performed to understand the interaction of gold surfaces with atomic H. For this purpose, we follow Norskov's^[28] idea in considering a three-atomic-layer thick Au(111) slab with a H atom adsorbed at three different sites, namely, the bridge, the face centered cubic (FCC), and the hexagonal close packed (HCP) site as illustrated in Figure 3a. The Au(111) surface was relaxed within the GGA scheme and no significant surface reconstruction was found.^[29] The potential energy (Figure 3b) was computed for the surface and bulk of the gold unit cell. Moreover, a global minimum is exhibited at a height of 1 Å from the center of the top layer for the bridge site, while the corresponding global minima for FCC and HCP sites lie at 0.8 Å. The calculated binding energy, $E_b = E(\text{AuH}) - E(\text{Au}) - E(\text{H})$, of hydrogen in all three configurations was found to be 2.30 eV. We have further found that this binding energy is in good accord with the state-of-the-art strongly constrained and appropriately normed (SCAN) meta-GGA functional.^[30] Such a value of binding energy is characteristic of chemisorption and is in agreement with previous results.^[31–33] Furthermore, our results are consistent with Ref.^[33] by Ferrin *et al.* which presents a comprehensive study of hydrogen adsorption, absorption and diffusion.

The prolonged robustness of the blue shift seen in Figure 2a over the 80–110 min region is most striking. The fact that the cell cover needed to be removed for the system to recover indicates that only under exposure to other environmental species one can lift H off the gold surface. This observation allows us to infer that H–Au bonding at the surface is as favored as the formation of H_2 . In fact, given that the binding energy of H_2 is 4.58 eV^[34] within GGA, the activation energy for desorption is $E_a = E(H_2) - E(2H)_{\text{surface}} = 20$ meV which shows that surface adsorption is preferred, thermal fluctuations notwithstanding. Therefore, these arguments justify the absence of immediate recovery as shown in Figure 2a indicating that a plasmonically driven gold hydride is formed and

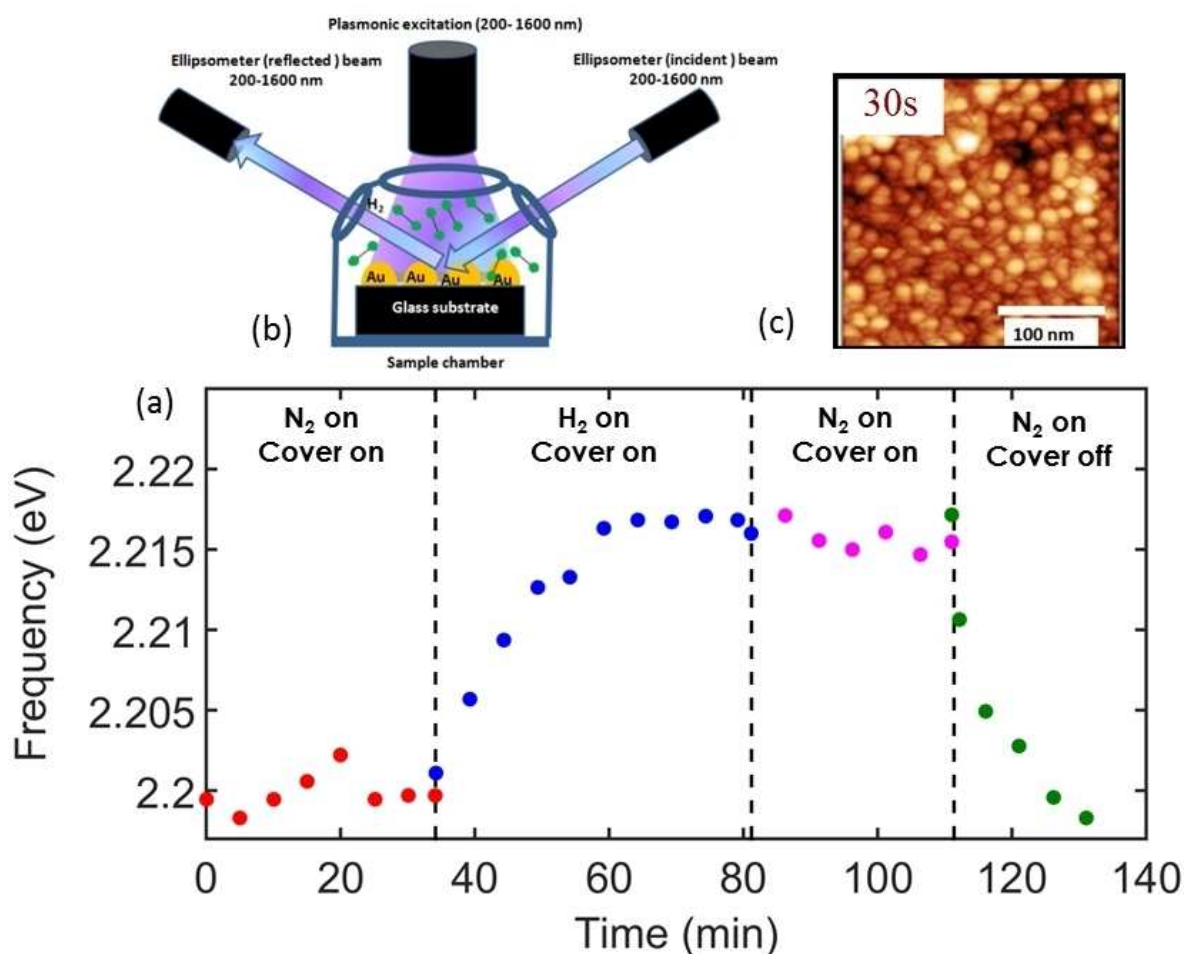


Figure 2. Monitoring changes in the peak of the imaginary part of the dielectric function under N₂ and H₂ exposures. (a) Time evolution of ϵ_2 peak location in frequency for gold nanoparticles. (b) Schematic of the *in situ* spectroscopic ellipsometry setup with broadband illumination from top. (c) AFM image of the 30 s sputtered dewetted gold nanoparticle thin-film sample showing particles of 10–20 nm size.

stabilized at the surface. Our results are also consistent with the desorption activation energies measured by Pan *et al.*^[35]

To further elucidate the nature of the bond between hydrogen and gold, Figure 4a shows the evolution of orbital occupation numbers of H as a function of the distance from the surface. A sharp transition is seen at approximately 2 Å bisecting two bonding regimes: covalent and van der Waals. In the covalent regime, at short distances from the gold surface, the populations of majority and minority occupied spin orbitals are equal and the electrons screening the proton have both *s* and *p* character. For distances greater than 2 Å, the screened proton essentially becomes a hydrogen atom interacting with the gold surface via van der Waals forces. The potential minimum occurs at around 0.8 Å before the hydrogen atom is formed, so that the associated bond has a covalent rather than a van der Waals character. The sticking of the hydrogen at the surface may also be interpreted as a proton attracted by the electrostatic image potential produced by the metallic surface. Moreover, our computations show that the proton on the 3-layer Au(111) surface leads to substantial charge displacements

and screening effects. These results are consistent with previous calculations by Takagi *et al.*^[31] and by Wang *et al.*^[32]

We simulated the effect of a thin gold hydride layer on the dielectric properties by placing an H atom at each of the three different available equilibrium sites. The dielectric function was obtained by calculating the imaginary part first as a joint-density-of-states with appropriate matrix elements. The real part was then obtained via a Kramers–Kronig transform. Dielectric function calculations were carried out using the Vienna *ab initio* simulation package (VASP).^[36–38] Results shown in Figure 5 correspond to 11% H coverage. The peak of the imaginary part of the ZZ-component of the dielectric tensor is seen to undergo a larger blue shift of 0.06 eV compared to the observed value of 0.02 eV, indicating that H-coverage in the simulations is larger than in the experimentally measured sample. We have verified within our calculations that the blue-shift as a function of the H coverage follows a linear relationship up to saturation, therefore the observed experimental shift of 0.02 eV corresponds to 4% theoretical H coverage. Furthermore, our blue shift is in good accord with Collins *et al.*^[27] and amplitude and lineshape of the calculated imaginary part of

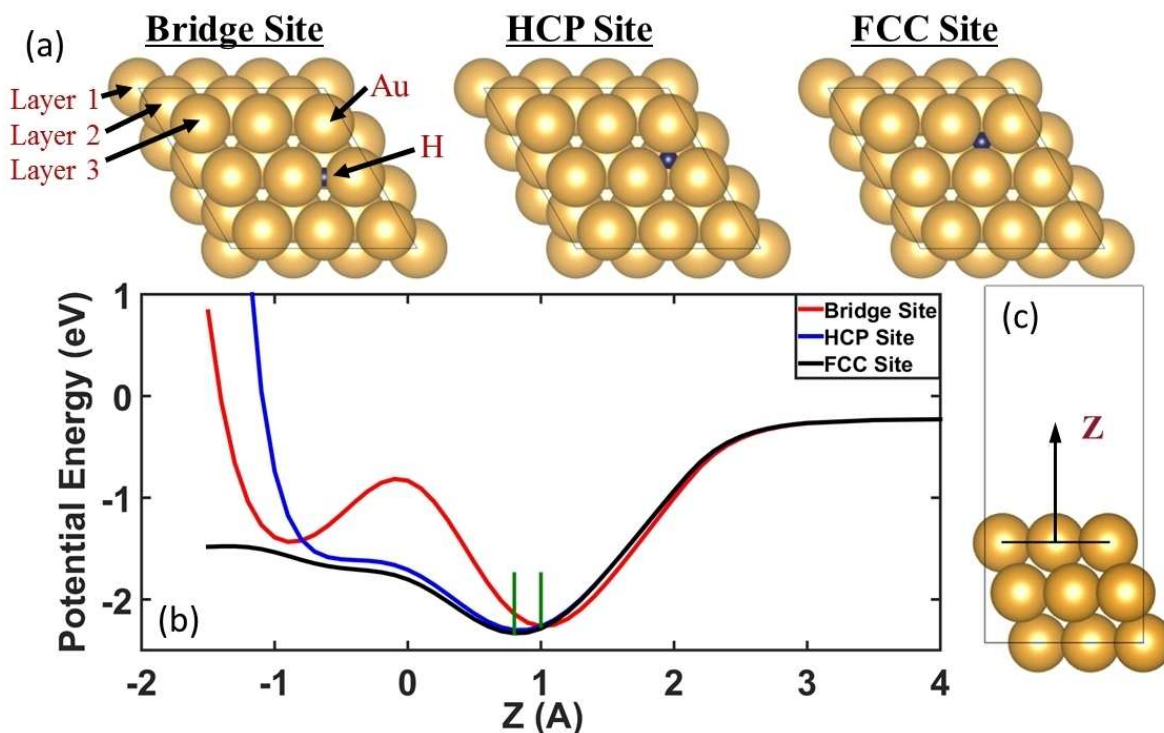


Figure 3. DFT calculations simulating the interaction of atomic hydrogen with gold under optical illumination. (a) Top view of three-atomic-layer Au(111) surface unit cell with a hydrogen atom (black) adsorbed at the bridge, hexagonal close packing, and face centered cubic sites, respectively. (b) Potential energy of a hydrogen atom for various adsorption sites as a function of the distance z measured from the center of the top gold layer (right). (c) Z-axis overlaid on the side view of the unit cell, with zero measured from the center of the top layer. For scale, note that the radius of the gold atoms is 1.47 Å.

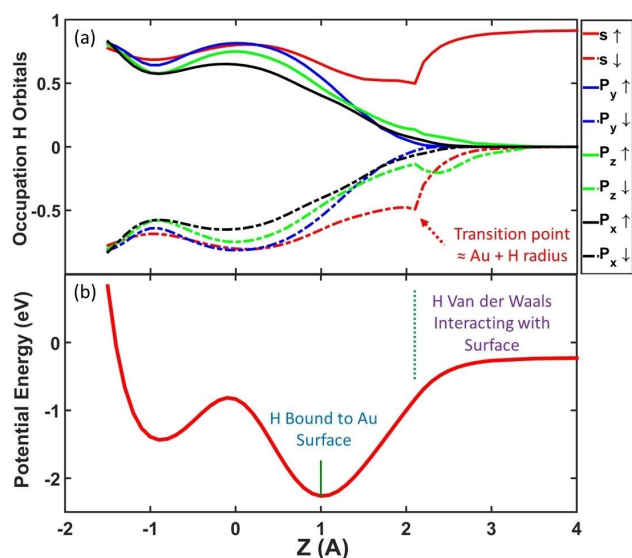


Figure 4. Different regimes in the interaction between the gold surface and the hydrogen atom. (a) H-atom orbital-projections as a function of height (Å) measured from the center of the top layer of the Au(111) surface. (b) Corresponding potential energy curve of hydrogen, where the transition at 2 Å is highlighted.

the dielectric function are also seen to agree remarkably well with the measurements of Giangregorio *et al.*^[2] Here, the

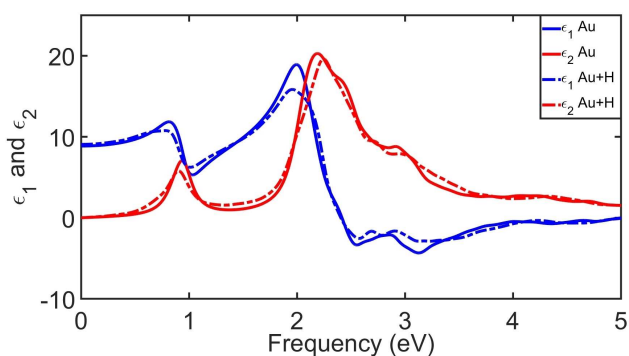


Figure 5. Theoretical dielectric function for Au(111) surface with and without hydrogen. Real (ϵ_1 , blue curves) and imaginary (ϵ_2 , red curves) parts of the dielectric function for bare Au(111) surface (solid) and Au(111)-H (dashed). An overall blue shift of 0.06 eV in the imaginary part of the dielectric function is seen.

differences with respect to Figure 1b are attributed to deviations in the nanoparticle thin films from an Au(111) surface.

We now discuss the robustness of our blue shift findings. Previous rationalizations of the blue shift utilize a model dielectric function focusing on the Drude part and a well-defined charge transfer.^[2,24,25,39] However, band structure contributions are vital for an accurate description of the optical properties of gold nano-structures^[40] and the AuH bond exhibits strong covalent character.^[41] Interestingly, small charge

transfer has been found in the case of hydrides^[41] while recent work on gold fluorides shows more significant charge transfer.^[42] In the present case, the screening of a proton by the electron gas is important. In fact, our calculations indicate that about 0.05 electron per AuH bond are displaced by Friedel oscillations near the gold surface. Nevertheless, these Friedel charge redistributions cannot be associated with charge transfer. Therefore, our blue-shift is not produced by charge transfer but by the formation of the hydride leading to an enhancement of the critical band gaps in the electronic structure.^[40] In conclusion, the correct blue-shift of the dielectric function can be explained only when full band structure and covalent bond formation are taken into account.

Conclusion

We have shown that solids can be transformed into new materials via plasmonic excitations. We demonstrate this novel pathway for creating new materials by considering the formation of a metastable gold hydride layer through the dissociation of molecular hydrogen on films of gold nanoparticles. This reaction also drives changes in the dielectric constant and the associated optical properties at the gold surface. Our study provides a new approach for exploring the bonding behavior between non-reactive noble metals and H₂ *in situ*, without the need for extreme experimental conditions,^[2,3] and opens up possibilities for novel methods of materials synthesis and low-concentration sensing applications. Clearly, our work shows that gold hydride should be considered as an intermediate in the crucial area of photocatalytic reactions such as the artificial photosynthesis of ammonia, which mitigates the high energy consumption of the Haber-Bosch process.^[43]

Supplementary Information Summary

Details pertaining to sample preparation, ellipsometry measurements and density functional theory calculations are given.

Author Contributions

DS, CL, EG contributed equally to this work.

DS, EG, SS, ZF, and EB contributed towards the design and implementation of ellipsometry measurements and the interpretation of the results. KG, MH, and SN were responsible for the thin film preparation. CL, BB, RM and AB were responsible for the theoretical calculations and the interpretation of the results. All contributed to the writing of the manuscript.

Acknowledgements

This work was supported by the Center for Complex Materials from First Principles (CCM), an Energy Frontier Research Center funded by the U.S. Department of Energy, Office of Science, Basic Energy Sciences under Award #DE-SC0012575 (thin film preparation, and computations applied to layered materials). The work at Northeastern University benefited from support of the US Depart-

ment of Energy (DOE), Office of Science, Basic Energy Sciences grant number DE-FG02-07ER46352, and Northeastern University's Advanced Scientific Computation Center and the National Energy Research Scientific Computing Center through DOE grant number DE-AC02-05CH11231. Ellipsometry measurements were funded by the NSF-Career grant (DMR-1350044), and partially by the startup funding from the University of Pennsylvania. Sample preparation was supported through a National Science Foundation Award (DMR-1707593).

Conflict of Interest

The authors declare no conflict of interest.

Keywords: Ab initio calculations · Ellipsometry · Gold · Hydride

- [1] S. Mukherjee, F. Libisch, N. Large, O. Neumann, L. V. Brown, J. Cheng, J. B. Lassiter, E. A. Carter, P. Nordlander, N. J. Halas, *Nano Lett.* **2013**, *13*, 240–247.
- [2] M. M. Giangregorio, M. Losurdo, G. V. Bianco, A. Operamolla, E. Dilonardo, A. Sacchetti, P. Capezzuto, F. Babudri, G. Bruno, *J. Phys. Chem. C* **2011**, *115*, 19520–19528.
- [3] V. F. Degtyareva, *Journal of Alloys and Compounds* **2015**, *645*, Supplement 1, S128–S131.
- [4] S. Linic, P. Christopher, D. B. Ingram, *Nat Mater* **2011**, *10*, 911–921.
- [5] I. Thomann, B. A. Pinaud, Z. Chen, B. M. Clemens, T. F. Jaramillo, M. L. Brongersma, *Nano Lett.* **2011**, *11*, 3440–3446.
- [6] S. Mubeen, J. Lee, N. Singh, S. Krämer, G. D. Stucky, M. Moskovits, *Nat Nano* **2013**, *8*, 247–251.
- [7] J. W. Schwede, I. Bargatin, D. C. Riley, B. E. Hardin, S. J. Rosenthal, Y. Sun, F. Schmitt, P. Pianetta, R. T. Howe, Z.-X. Shen, *Nat Mater* **2010**, *9*, 762–767.
- [8] J. R. Renzas, G. A. Somorjai, *J. Phys. Chem. C* **2010**, *114*, 17660–17664.
- [9] N. A. Joy, B. K. Janiszewski, S. Novak, T. W. Johnson, S.-H. Oh, A. Raghunathan, J. Hartley, M. A. Carpenter, *J. Phys. Chem. C* **2013**, *117*, 11718–11724.
- [10] M. Maillard, P. Huang, L. Brus, *Nano Lett.* **2003**, *3*, 1611–1615.
- [11] Y. Zhai, J. S. DuChene, Y.-C. Wang, J. Qiu, A. C. Johnston-Peck, B. You, W. Guo, B. DiCiaccio, K. Qian, E. W. Zhao, *Nat Mater* **2016**, *15*, 889–895.
- [12] M. Murdoch, G. I. N. Waterhouse, M. A. Nadeem, J. B. Metson, M. A. Keane, R. F. Howe, J. Llorca, H. Idriss, *Nat Chem* **2011**, *3*, 489–492.
- [13] W. Hou, W. H. Hung, P. Pavaskar, A. Goeppert, M. Aykol, S. B. Cronin, *ACS Catal.* **2011**, *1*, 929–936.
- [14] D. Conklin, S. Nanayakkara, T.-H. Park, M. F. Lagadec, J. T. Stecher, X. Chen, M. J. Therien, D. A. Bonnell, *ACS Nano* **2013**, *7*, 4479–4486.
- [15] F. Wu, L. Tian, R. Kanjolia, S. Singamaneni, P. Banerjee, *ACS Appl. Mater. Interfaces* **2013**, *5*, 7693–7697.
- [16] Z. Gueroui, A. Libchaber, *Phys. Rev. Lett.* **2004**, *93*, 166108.
- [17] L. Hayati, C. Lane, B. Barbiellini, A. Bansil, H. Mosallaei, *Phys. Rev. B* **2016**, *93*, 245411.
- [18] B. Barbiellini, S. Das, V. Renugopalakrishnan, P. Somasundaran, *Condensed Matter* **2018**, *3*, 10.
- [19] C. Li, Y. Zhang, M. T. Cole, S. G. Shivareddy, J. S. Barnard, W. Lei, B. Wang, D. Pribat, G. A. J. Amaratunga, W. I. Milne, *ACS Nano* **2012**, *6*, 3236–3242.
- [20] D. Sil, K. D. Gilroy, A. Niaux, A. Boulesbaa, S. Neretina, E. Borguet, *ACS Nano* **2014**, *8*, 7755–7762.
- [21] K. D. Gilroy, A. Sundar, M. Hajfathalian, A. Yaghouzade, T. Tan, D. Sil, E. Borguet, R. A. Hughes, S. Neretina, *Nanoscale* **2015**, *7*, 6827–6835.
- [22] I. P. Silverwood, S. M. Rogers, S. K. Callar, S. F. Parker, C. R. A. Catlow, *Chem. Commun.* **2016**, *52*, 533–536.
- [23] T. Whittaker, K. B. S. Kumar, C. Peterson, M. N. Pollock, L. C. Grabow, B. D. Chandler, *J. Am. Chem. Soc.* **2018**, *140*, 16469–16487.
- [24] R. Ishida, S. Yamazoe, K. Koyasu, T. Tsukuda, *Nanoscale* **2016**, *8*, 2544–2547.
- [25] R. Ishida, S. Hayashi, S. Yamazoe, K. Kato, T. Tsukuda, *J. Phys. Chem. Lett.* **2017**, *8*, 2368–2372.

- [26] D. Sil, *Synthesis and Applications of Plasmonic Nanostructures*, Temple University, **2015**.
- [27] S. S. E. Collins, M. Cittadini, C. Pecharromán, A. Martucci, P. Mulvaney, *ACS Nano* **2015**, *9*, 7846–7856.
- [28] B. Hammer, J. K. Nørskov, in (Ed.: B.-A. in Catalysis), Academic Press, **2000**, pp. 71–129.
- [29] P. L. Hansen, J. B. Wagner, S. Helveg, J. R. Rostrup-Nielsen, B. S. Clausen, H. Topsøe, *Science* **2002**, *295*, 2053–2055.
- [30] J. Sun, A. Ruzsinszky, J. P. Perdew, *Phys. Rev. Lett.* **2015**, *115*, 036402.
- [31] S. Takagi, J. Hoshino, H. Tomono, K. Tsumuraya, *J. Phys. Soc. Jpn.* **2008**, *77*, 054705.
- [32] S.-W. Wang, W. H. Weinberg, *Surface Science* **1978**, *77*, 14–28.
- [33] P. Ferrin, S. Kandoi, A. U. Nilekar, M. Mavrikakis, *Surface Science* **2012**, *606*, 679–689.
- [34] G. Kresse, J. Hafner, *Surface Science* **2000**, *459*, 287–302.
- [35] M. Pan, A. J. Brush, Z. D. Pozun, H. Chul Ham, W.-Y. Yu, G. Henkelman, G. S. Hwang, C. Buddie Mullins, *Chemical Society Reviews* **2013**, *42*, 5002–5013.
- [36] G. Kresse, J. Furthmüller, *Computational Materials Science* **1996**, *6*, 15–50.
- [37] G. Kresse, J. Furthmüller, *Phys. Rev. B* **1996**, *54*, 11169–11186.
- [38] M. Gajdoš, K. Hummer, G. Kresse, J. Furthmüller, F. Bechstedt, *Phys. Rev. B* **2006**, *73*, 045112.
- [39] W. L. Watkins, Y. Borensztein, *Phys. Chem. Chem. Phys.* **2017**, *19*, 27397–27405.
- [40] D. Rioux, S. Vallières, S. Besner, P. Muñoz, E. Mazur, M. Meunier, *Advanced Optical Materials* **2014**, *2*, 176–182.
- [41] P. Pyykkö, *Angewandte Chemie International Edition* **2004**, *43*, 4412–4456.
- [42] J. Lin, S. Zhang, W. Guan, G. Yang, Y. Ma, *J. Am. Chem. Soc.* **2018**, *140*, 9545–9550.
- [43] T. Oshikiri, K. Ueno, H. Misawa, *Angew. Chem. Int. Ed.* **2016**, *55*, 3942–3946.

Submitted: March 11, 2019

Accepted: March 19, 2019